

Zhou Pei-Yuan Centre for Applied Mathematics, Tsinghua University November 2013

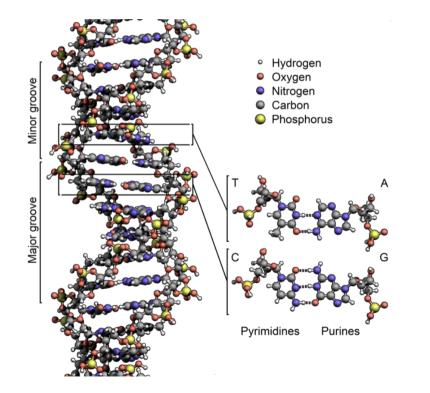
F. Piazza Center for Molecular Biophysics and University of Orléans, France

Selected topic in Physical Biology Lecture 2

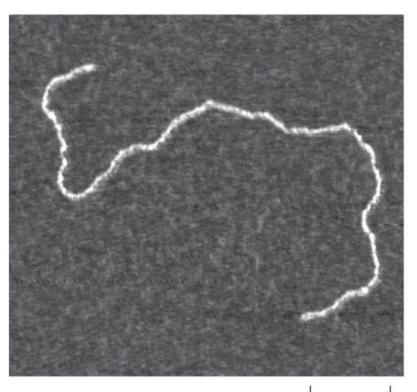
The structure of biological macromolecules: Introduction to statistical models of polymers and applications to DNA stretching

Deterministic versus statistical description of structure

Coordinate (PDB) files reflect a **deterministic** description of macromolecular structure. In a sense it is a deceptively static picture that can only thought of as a starting point



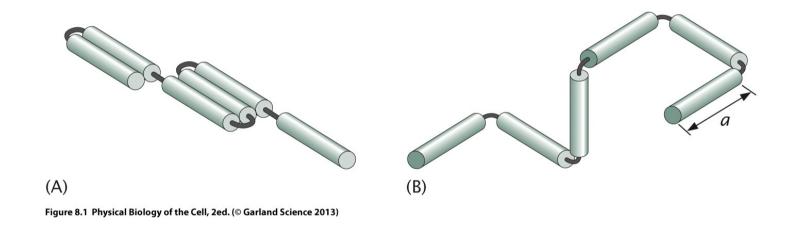
Notion of universality comes immediately when thinking to polymeric systems (e.g. DNA). Think of a virus DNA ejected into the cell, relevant questions are (i) how much space it takes (ii) where in the cell it does so





The random walk model

Deterministic: vector r(s) of position at a distance s along the contour Statistical: Rigid segments of length a connected by flexible links



1D: segments at +/-180^o with respect to each other

3D: where links are restricted to 90⁰ angles

1D random walk model

The history of the walker is built as a sequence of N right and left moves, each with probability 1/2. Each move starts anew (no correlation).

 \circ There are a total of 2^{*N*} admissible configurations \circ All configurations are equally probable (1/2^{*N*})

The mean size of the polymer scales as the square root of the number of segments ("diffusion")

The i-th step of the walk is $x_i = \pm a$

$$\langle R \rangle = \left\langle \sum_{i=1}^{N} x_i \right\rangle = 0$$
$$\langle R^2 \rangle = \left\langle \sum_{i,j=1}^{N} x_i x_j \right\rangle = \sum_{i=1}^{N} \langle x_i^2 \rangle + \sum_{i \neq j}^{N} \langle x_i x_j \rangle = Na^2$$

$$\sqrt{\langle R^2 \rangle} = \sqrt{N}a$$

Probability of a given macromolecular state

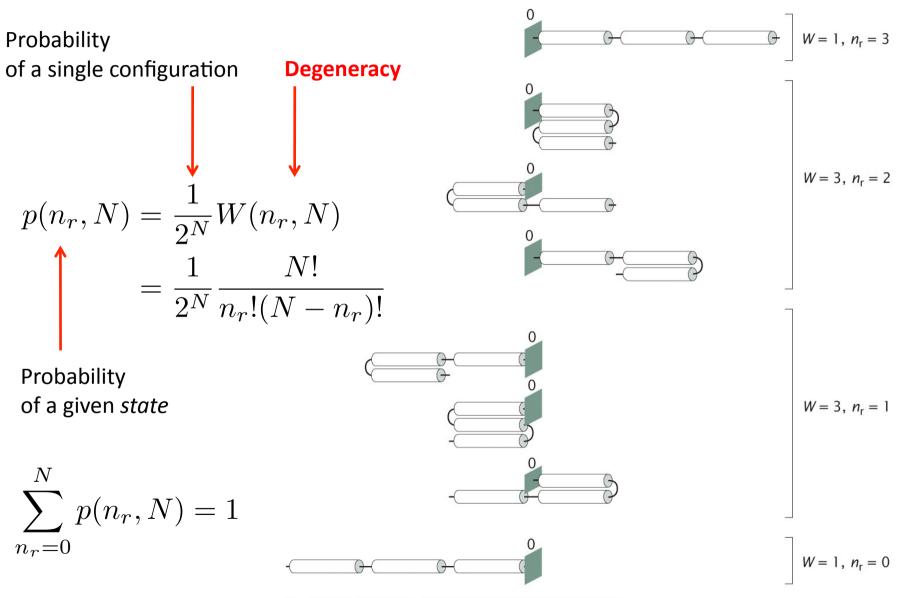


Figure 8.3 Physical Biology of the Cell, 2ed. (© Garland Science 2013)

Probability distribution of end-to-end distance

$$(n_r - n_l)a = R$$

$$n_r + n_l = N$$

$$p(n_r, N) \rightarrow p(R, N) = \frac{N!}{\left(\frac{N}{2} + \frac{R}{2a}\right)! \left(\frac{N}{2} - \frac{R}{2a}\right)!} \left(\frac{1}{2}\right)^N$$

Use Stirling approximation and expand the logarithms...

$$\log(N!) \approx N \log N - N + \log \sqrt{2\pi N}$$
$$\log(1 \pm x) = \pm x - \frac{x^2}{2} + \mathcal{O}(x^3)$$

After some algebra
$$\longrightarrow \log p(R, N) = \log 2 - \log \sqrt{2\pi N} - \frac{R^2}{2Na^2}$$
 (do it!)

$$P(R,N) = \frac{p(r,N)}{2a} = \frac{1}{\sqrt{2\pi Na^2}} e^{-R^2/2Na^2}$$

Overwhelming probability of zero separation of ends

It is nothing but the **central limit theorem**

$$R = \sum_{i} x_i$$

is Gaussian-distributed, if xi are iid variables with finite mean and variance

... and it is Gaussian in three dimensions as well!

It immediately follows that the distribution of end-to-end separation is Gaussian in three dimensions as well (the conditions of the central limit theorem are met)

The persistence length

The persistence length is a measure of the length scale over which a polymer remains roughly straight



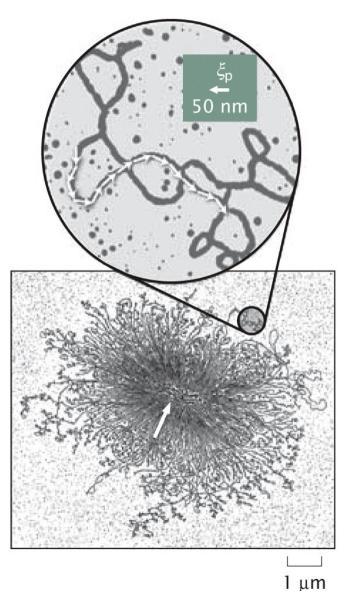
Tangent vector at a position s along the contour length

Persistence length

$$\langle \mathbf{t}(s) \cdot \mathbf{t}(u) \rangle = e^{-|s-u|/\xi_p}$$
Average over all the configurations

This formula is valid for a free polymer. If the polymer is subject to constraints (i.e. ends attached) the correlation will change

Bacterial genome that has escaped the bacterial cell



The Kuhn length

A good example of long flexible polymer is provided by genomic DNA of viruses such as λ -phage, with a contour length of $L = 16.6 \mu m$.

This should be compared with the persistence length of DNA, ξ_p about 50 nm (room temperature and solvent conditions typical of the cellular environment).

By the very definition of persistence length, we can think of the **polymer as consisting** of *L* / ξ_p connected links that take random orientations with respect to each other. This is the logic that gives rise to the **freely-jointed chain** model

The length of the uncorrelated steps is known as Kuhn length

Let us derive the relation between the Kuhn length and the persistence length. We start from the end-to-end vector

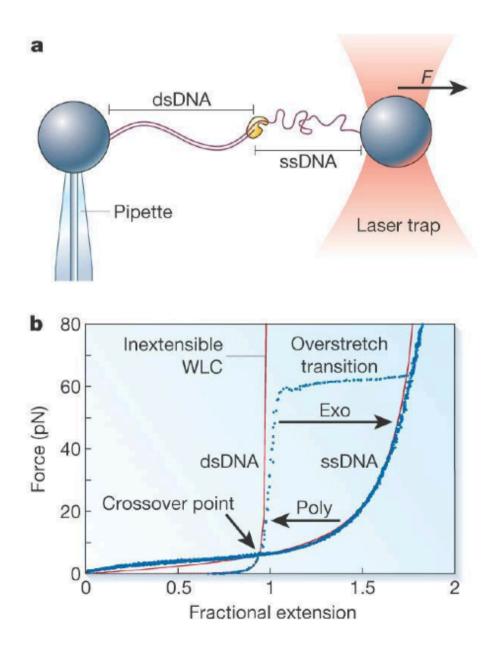
$$\mathbf{R} = \sum_{i=1}^{N-1} (\mathbf{x}_{i+1} - \mathbf{x}_i) \Longrightarrow \int_0^L \mathbf{t}(s) \, ds$$

The Kuhn length and the persistence length

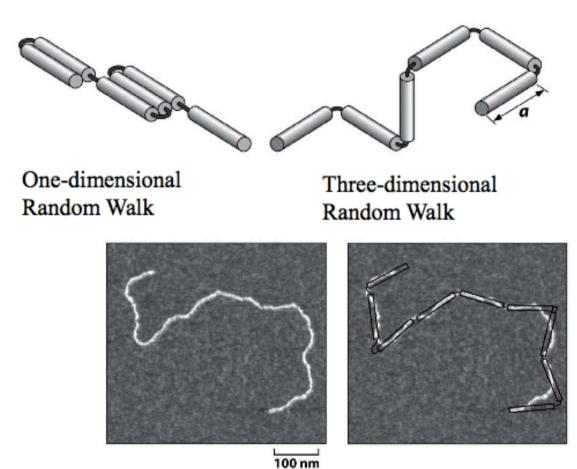
$$\begin{split} \langle \mathbf{R} \cdot \mathbf{R} \rangle &= \int_0^L ds \int_0^L du \, \langle \mathbf{t}(s) \cdot \mathbf{t}(u) \rangle \\ &= \int_0^L ds \int_0^L du \, e^{-|s-u|/\xi_p} \approx 2L\xi_p \qquad L \gg \xi_p \end{split}$$
 From the random walk model we know that
$$\langle \mathbf{R} \cdot \mathbf{R} \rangle = Na^2 = La \Longrightarrow a = 2\xi_p \end{split}$$

This gives a quantitative rule that allows to treat a given polymer with a given persistence length as a random walk (freely-jointed chain)

Interpreting force-extension curves of DNA



Freely-jointed chain model



- Macromolecules are viewed as rigid segments connected by hinges (hypothetical).
- N bonds, N+1 rigid monomers
- All rigid segments have equal length.
- Every macromolecular configuration is equally probable.
- The molecule can sample many accessible configurations, all close in energy relative to kT.

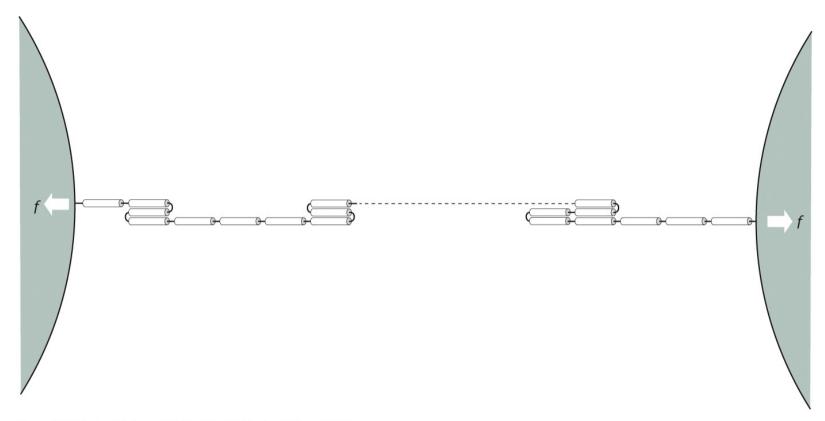
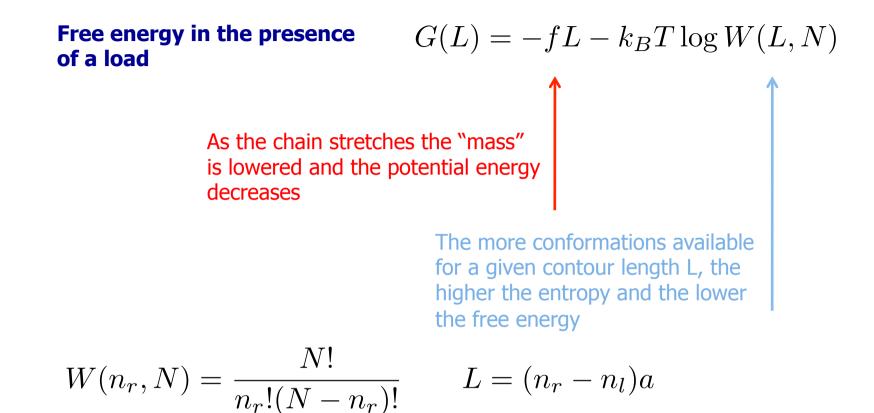


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The dangling mass analogy of a constant load



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Using Stirling approximation and neglecting constant terms (only free-energy differences matter!)

$$G(L) = -2fa n_r + k_B T [n_r \log n_r + (N - n_r) \log(N - n_r)]$$

The most probable value of *L* minimizes the free energy

$$\frac{\partial G(L)}{\partial n_r} = 0 \Longrightarrow \begin{cases} n_r = \frac{N}{1 + e^{-2fa/k_BT}} \\ n_l = \frac{Ne^{-2fa/k_BT}}{1 + e^{-2fa/k_BT}} \end{cases}$$

$$L = (n_r - n_l)a = Na \tanh\left(\frac{fa}{k_B T}\right)$$

In the small-force regime, the DNA behaves as an <u>entropic spring</u>

$$fa \ll k_B T \Longrightarrow \begin{cases} \text{Double-stranded DNA, } a \approx 100 \text{ nm} \Longrightarrow f \ll 40 \text{ fN} \\ \text{Single-stranded DNA, } a \approx 1.5 \text{ nm} \Longrightarrow f \ll 3 \text{ pN} \end{cases}$$

In three dimensions ...

The calculations are simpler if we assume that at each site the monomer vector can assume all orientations.

The one-particle partition function

$$Z_{1} = \int_{0}^{2\pi} d\varphi \int_{0}^{\pi} e^{fa\cos\vartheta/k_{B}T} \sin\vartheta \,d\vartheta$$
$$= 2\pi \int_{-1}^{1} e^{fax/k_{B}T} \,dx$$
$$= 4\pi \left(\frac{k_{B}T}{fa}\right) \sinh\left(\frac{fa}{k_{B}T}\right)$$

The N-particle partition function

$$Z = (Z_1)^N$$

Note that the factor 1/N! is absent here. The reason is that the monomers are not undistinguishable and the polymer is virtually *directed* (ends fixed)

$$G = -k_B T \log Z = -k_B T \left\{ \log \left[\sinh \left(\frac{fa}{k_B T} \right) \right] - \log \left[\frac{fa}{k_B T} \right] \right\}$$

The average chain length is the thermodynamically conjugate variable of the external force f

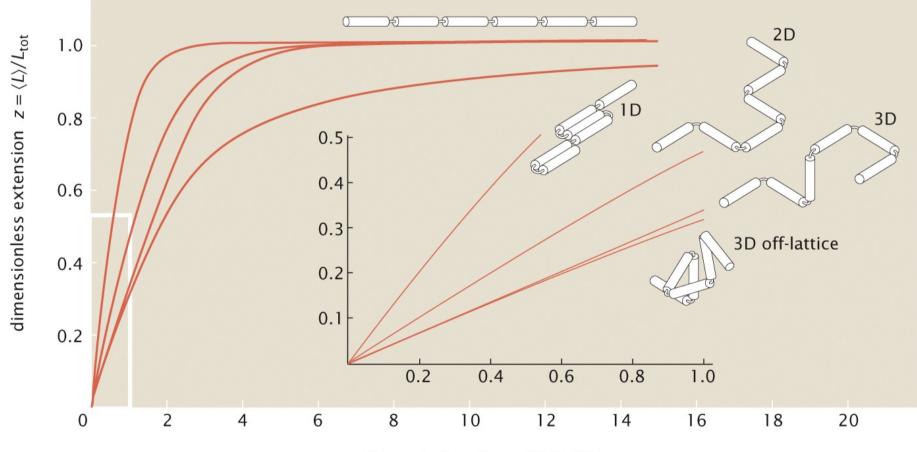
$$\langle L \rangle = -\frac{\partial G}{\partial f} = Na \left[\coth\left(\frac{fa}{k_B T}\right) - \frac{k_B T}{fa} \right]$$

$$\operatorname{coth} \simeq \frac{x}{3} + \frac{1}{x} \quad \text{as } x \to 0$$

$$\longrightarrow \quad \frac{\langle L \rangle}{N} \simeq \left(\frac{a}{3k_BT}\right) f$$

Again <u>entropic spring</u>. The force constant is three times stiffer than in 1D

$$\frac{\langle L \rangle}{L_{\text{tot}}} = \begin{cases} \tanh\left(\frac{fa}{k_BT}\right) \simeq \left(\frac{a}{k_BT}\right) f & 1D\\ \coth\left(\frac{fa}{k_BT}\right) - \frac{k_BT}{fa} \simeq \left(\frac{a}{3k_BT}\right) f & 3D \end{cases}$$



dimensionless force $(fa/k_{\rm B}T)$

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What do the experiment reveal?

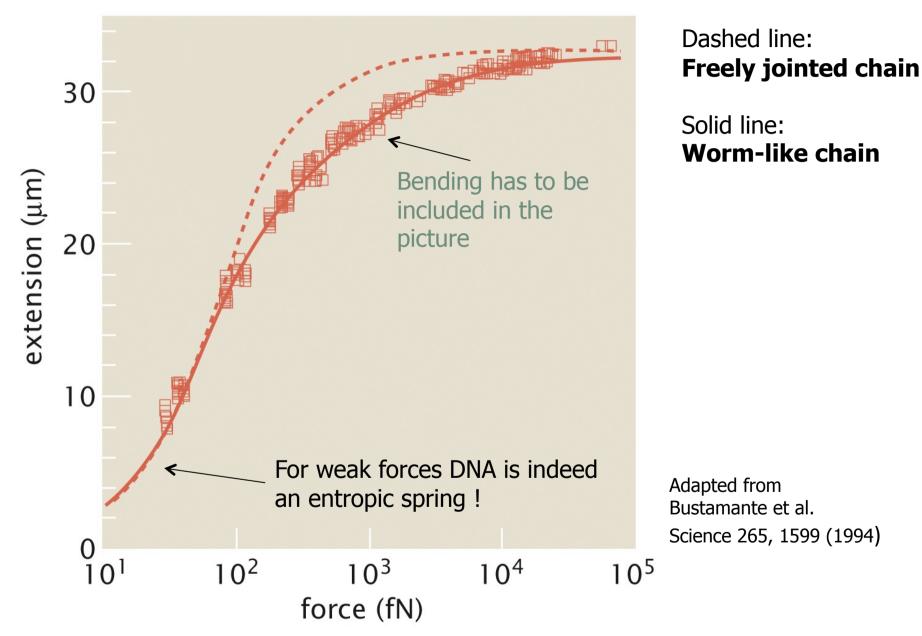


Figure 5.14 Physical Biology of the Cell, 2ed. (© Garland Science 2013)